

## PHASE STUDIES IN THE SYSTEMS $\text{Ag}_2\text{Te}-\text{Ga}_2\text{Te}_3$ , $\text{ZnSe}-\text{In}_2\text{Se}_3$ , AND $\text{ZnS}-\text{Ga}_2\text{S}_3$

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### SUMMARY

The title systems were investigated by means of differential thermal analysis (DTA) and X-ray phase analysis (XPA). From these measurements phase diagrams were constructed; crystal data of intermediate compounds and a structure model of  $\text{ZnGa}_2\text{S}_4$  are added.

### INTRODUCTION

Ternary semiconducting crystals with the chalcopyrite and defect-chalcopyrite structure find growing interest because of their non-linear optical properties. As a suitable basis for their crystal growth, the phase relations in the corresponding systems should be known. For this reason, the systems  $\text{Ag}_2\text{Te}-\text{Ga}_2\text{Te}_3$ ,  $\text{ZnSe}-\text{In}_2\text{Se}_3$ , and  $\text{ZnS}-\text{Ga}_2\text{S}_3$  were (re-)examined and their tentative phase diagrams as well as the results of X-ray diffraction studies on some of the intermediate compounds are dealt with.

### EXPERIMENTAL

The binary components were synthesized from high purity elements. DTA experiments were performed and evaluated as described earlier [1]. X-ray phase analyses of the DTA products were done with a Guinier-de Wolff camera ( $\text{CuK}\alpha$  radiation) using cubic  $\text{As}_2\text{O}_3$  as internal standard and programs GUINIER [2] and LSUCRE [3] for precise determination of lattice parameters. Single-crystals were prepared by chemical vapour transport (CVT). For a structure analysis of  $\text{ZnGa}_2\text{S}_4$ , an intensity data set of a suitable CVT crystal was collected on an ENRAF-NONIUS CAD-4 diffractometer (graphite-monochromatized  $\text{AgK}\alpha$  radiation). Calculations were performed with the X-RAY76 program system [4] on the basis of 857 unique observed reflections, corrected for absorption and extinction effects.

## RESULTS

1. The system  $\text{Ag}_2\text{Te}-\text{Ga}_2\text{Te}_3$ .

Palatnik and Belova [5] studied this system in the range 20-100 mole %  $\text{Ga}_2\text{Te}_3$ . A complete re-examination of the phase diagram over the whole concentration region (Fig. 1) shows no essential differences of the already published phase boundaries, but at the  $\text{Ag}_2\text{Te}$  rich side of the system, a new compound of composition  $\text{Ag}_9\text{GaTe}_8$  was found. It melts incongruently at 705(5) °C and shows a phase transition in the temperature interval 2-30 °C. Contrary to the homologous compounds in the systems  $\text{Ag}_2\text{S}-\text{Ga}_2\text{S}_3$  [6] and  $\text{Ag}_2\text{Se}-\text{Ga}_2\text{Se}_3$  [7-9] which belong to a large family of isotypic compounds (the "argyrodites", space group  $F\bar{4}3m$  of the aristotypes [10]) this compound crystallizes hexagonally [11]. Crystal data of both polymorphs of  $\text{Ag}_9\text{GaTe}_8$  and of  $\text{AgGaTe}_2$  (chalcopyrite structure type) are listed in Table 1. So far, the Ga rich phase  $\text{AgGa}_3\text{Te}_8$  could not be obtained single-crystalline.

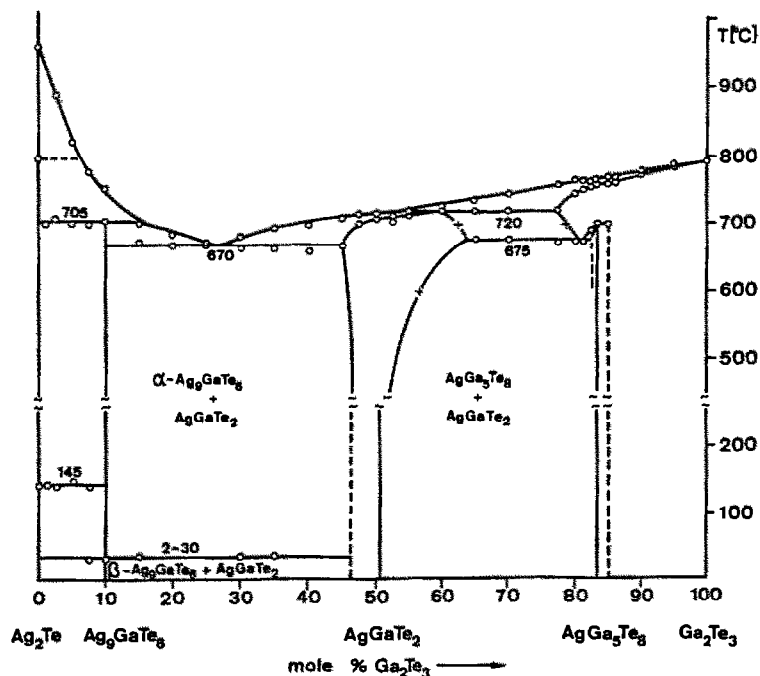


Fig. 1. Phase diagram of the system  $\text{Ag}_2\text{Te}-\text{Ga}_2\text{Te}_3$ .

## 2. The system ZnSe-In<sub>2</sub>Se<sub>3</sub>.

Fig. 2 shows the proposed phase diagram; the only intermediate compound found, ZnIn<sub>2</sub>Se<sub>4</sub>, melts incongruently at 960(10) °C. Small DTA effects and electrical conductivity measurements indicate a phase transition at 45(5)°C. A structure analysis [12] showed that ZnIn<sub>2</sub>Se<sub>4</sub> crystallizes with space group I4̄2m, i.e. a variety of the thiogallate/defect-chalcopyrite structure, caused by statistical disorder of the cations. Its crystal data are listed in Table 1.

Hauseleer et al. [13] report on a second intermediate compound of composition Zn<sub>0.4</sub>In<sub>2</sub>Se<sub>3.4</sub> ( $\approx$  2ZnSe.5In<sub>2</sub>Se<sub>3</sub>) which cannot be confirmed unequivocally: the continuously detected  $\alpha$ - $\beta$  phase transition of In<sub>2</sub>Se<sub>3</sub> suggests a two-phase region between ZnIn<sub>2</sub>Se<sub>4</sub> and In<sub>2</sub>Se<sub>3</sub>, but weak additional X-ray diffraction lines point at ambiguous equilibrium conditions in the related composition/temperature region.

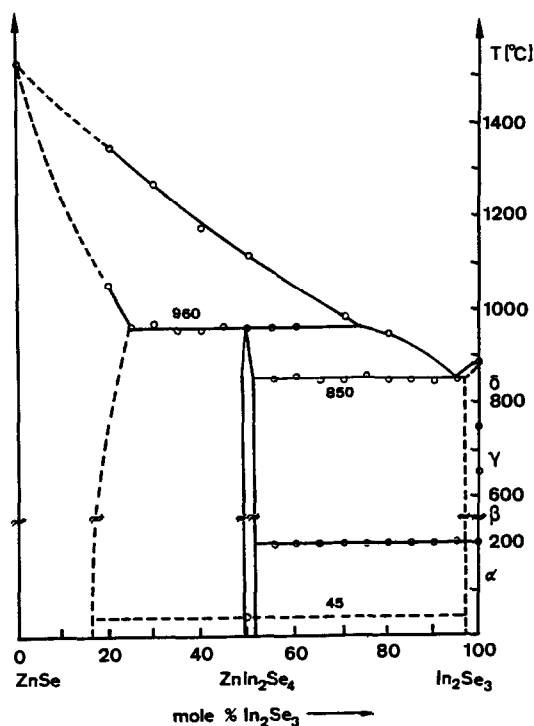


Fig. 2: Phase diagram of the system ZnSe-In<sub>2</sub>Se<sub>3</sub>.

### 3. The system $\text{ZnS-Ga}_2\text{S}_3$ .

Rather preliminary and contradictory phase diagrams were published by Flahaut et al. [14] and Malevskii [15]. Fig. 3 shows the results of our treatment. At high temperatures, a complete solid solution between  $\text{ZnS}$  and  $\text{Ga}_2\text{S}_3$  with defect-wurtzite structure exists as confirmed by quenching annealed specimens from temperatures just below the solidus. By DTA the liquidus and solidus curves could be established up to  $1300^\circ\text{C}$ , but in the subsolidus region considerable kinetic barriers prevent the elucidation of the phase boundaries. Besides  $\text{ZnGa}_2\text{S}_4$  (its crystal data shows Tab. 1), a second phase with a phase width from 80 to 92 mole %  $\text{Ga}_2\text{S}_3$  was found; various attempts to prepare single-crystals failed.

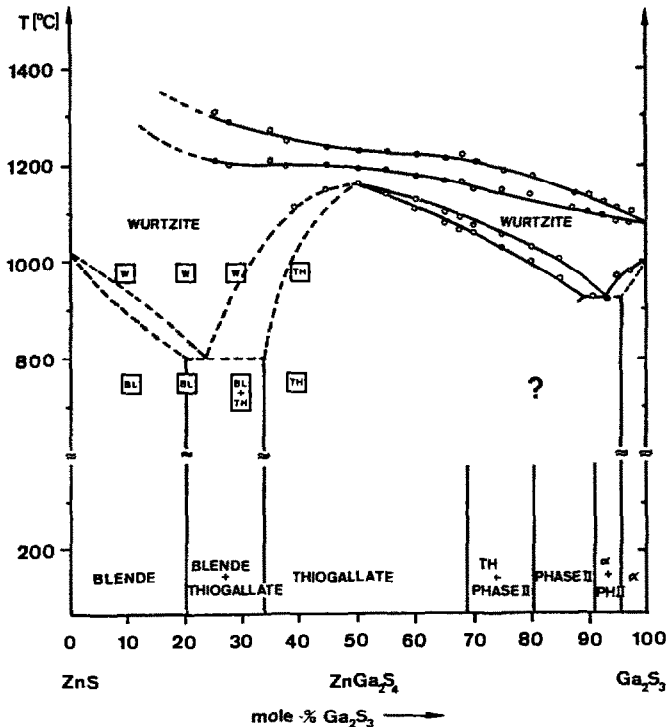


Fig. 3. Tentative phase diagram of the system  $\text{ZnS-Ga}_2\text{S}_3$ .

Phases in boxes were identified after annealing and quenching (TH: thio-gallate, W: wurtzite).

TABLE 1

Crystal data.

		Ag <sub>3</sub> BaTe <sub>4</sub> a [111]	Ag <sub>3</sub> BaTe <sub>4</sub> B [111]	AgBaTe <sub>2</sub>	ZnIn <sub>2</sub> S <sub>4</sub> [12]	ZnBa <sub>2</sub> S <sub>4</sub>
a	(Å)	8.260(5)	16.498(4)	6.313(2)	5.7093(7)	5.290(1)
c	(Å)	13.407(8)	13.274(5)	11.981(6)	11.449(2)	10.354(5)
c/a		1.623	0.805	1.898	2.005	1.957
V	(Å <sup>3</sup> )	792.2	3128.9	477.5	372.2	289.8
Z		2	8	4	2	2
D <sub>x</sub>	(gcm <sup>-3</sup> )	7.57	7.67	6.02	5.43	3.70
D <sub>o</sub>	(gcm <sup>-3</sup> )				5.51	3.76
SB		P6 <sub>3</sub> /mcc	P6 <sub>3</sub> /m	I4̄2d	I4̄2m	I4̄
Habit		< plates >		bisphenoids	columns	bisphenoids
Form		{001}, {100}, {101}		{112}	{112}, {001}	{112}
Colour		< black >		black	dark red	colourless

#### 4. Structure of ZnGa<sub>2</sub>S<sub>4</sub>.

Zn and Ga cannot be discriminated in X-ray experiments because of their nearly identical scattering factors. However, Haeuseler [16] and Siebert [17] postulated on the basis of FIR, Raman, and spinprobe investigations that ZnGa<sub>2</sub>S<sub>4</sub> possesses an ordered arrangement of the Zn and Ga atoms in accordance with the thiogallate structure type (prototype CdGa<sub>2</sub>S<sub>4</sub> [18]). Therefore, the structure of ZnGa<sub>2</sub>S<sub>4</sub> was treated in space group I4̄ with starting parameters from [18] and substituting Cd by Zn; refinement converged at R=4.5%. Tables 2 and 3 show the results of the structure analysis.

TABLE 2

Fractional atomic and anisotropic thermal (Å<sup>2</sup>x10<sup>2</sup>) parameters, e.s.d.'s in parentheses.

Atom	x	y	z	U <sub>11</sub>	U <sub>22</sub>	U <sub>33</sub>	U <sub>12</sub>	U <sub>13</sub>	U <sub>23</sub>
Me1 (Zn)	0	0	0	1.94(7)	1.94(7)	1.6(1)	-	-	-
Me2 (Ba)	0	0	0.5	0.68(4)	0.68(4)	0.8(1)	-	-	-
Me3 (Ba)	0	0.5	0.25	1.39(4)	1.39(4)	1.02(7)	-	-	-
B	0.2541(4)	0.2418(4)	0.1314(2)	1.53(7)	1.53(7)	1.12(6)	-0.14(6)	0.35(5)	0.10(3)

TABLE 3

Bond distances (Å).

Me1 (Zn) - S	2.301(2)	4x
Me2 (Ga) - S	2.328(2)	4x
Me3 (Ga) - S	2.276(2)	4x

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